







AN ANALYSIS OF INTERLAMINAR STRESS GRADIENTS AND IMPACT DAMAGE IN GRAPHITE-EPOXY LAMINATES

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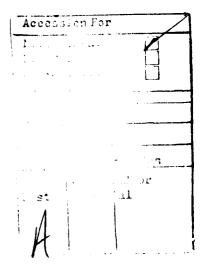
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LIST OF SYMBOLS

| <u>Letter</u> | <u>Definition</u> |
|---------------------------------|---|
| (A _n) _{ij} | Integrated ply property moments; $n = 0,1,2$. |
| B i a | Composite stress-curvature stiffness constants from classical 3D lamination theory. |
| Č _{ij} | Composite stress-strain stiffness constants from classical 3D lamination theory. |
| C [*] ij(ξ) | Composite stress-strain parametric stiffness coefficients for 3D finite element laminate modeling. |
| CCL, etc. | Three character mnemonic defining 3D finite elements where $C = \text{cubic}$, $L = \text{linear}$ indicate displacements in the coordinate directions associated with the character position. |
| $\bar{D}_{\alpha\beta}$ | Composite moment-curvature stiffness constants from classical 3D lamination theory. |
| e ~i | Orthonormal Cartesian bases vectors. |
| E _{ij} | Orthotropic material extensional moduli. |
| F ₁ (ξ) | Parametric cubic Hermite functions. |
| G _{ij} | Orthotropic material shear moduli. |
| P ijk | Tricubic Hermite coefficients in point format. |
| R,r | Radial coordinate. |
| $S_{ij}^{\star}(\xi)$ | Composite strain-stress parametric flexibility coefficients for 3D finite element laminate model. |
| S _m | Cubic Hermite coefficients in algebraic format. |
| T _{mn} | Point format cubic transformation matrix. |
| Z _i | Cartesian coordinate. |
| Greek Letter | Definition |
| δ _i | Coefficients in the solution of the contact problem for transversely isotropic materials. |
| ϵ_{i}^{\star} | Composite strain component. |
| κ _α | Laminate curvatures in the surface coordinate directions of the composite material. |

| <u>Greek Letter</u> | <u>Definition</u> |
|-------------------------------|---|
| ν _{ij} | Orthotropic material Poisson ratios. |
| $\sigma_{\mathbf{i}}^{\star}$ | Composite stress component. |
| ξį | Parametric coordinate $0 \le \xi_i \le 1$. |

INTRODUCTION AND SUMMARY

The sensitivity of laminates to low velocity impact damage is a practical design problem that the ASTM and others have been investigating for several years. One of the difficulties in studying impact response is predicting the complex interlaminar stress gradients that it produces in composite structures. This report presents a systematic approach to interlaminar stress gradient modeling with applications to a graphite-epoxy laminate impacted by a steel sphere and a laminate with an internal disbond. The approach is based on a family of variable property finite elements for laminate models that transition to discrete ply models for regions with interlaminar stress gradients. All 45 elastic constants for a general anisotropic laminate are simulated and considerable error is shown to occur when uniform properties are used with certain laminate models. Careful modeling of the force-deformation behavior is required to predict accurate boundary conditions in the impact region when bending is present. Numerical results are presented for a thin 8 ply laminate and a thick 48 ply laminate.

The report first develops two new finite element modeling tools for interlaminar stress analysis and validates them using control problems with known solutions. The new analysis tools are composite property models that vary through the thickness to correctly model bending and linear element constraints for lamination theory force-deformation modeling. These tools are then used to analyze a thin laminate with an internal disbond and a thick laminate impacted at low velocity by a steel sphere. The disbond produces a locally unbalanced laminate that under biaxial compression warps badly with large out-of-plane displacements, but with very little internal force redistribution. These results were obtained from a model using two planes of symmetry that were found to introduce ficticious constraint forces between 0 and 45 degree plies at the laminate center. Results for the laminate impact problem were obtained from a complete 360 degree model of the impact site. A structural model of the entire laminate was used to obtain displacement boundary conditions for a discrete model of the impact site. These results show large subsurface shears at the perimeter of the contact area in the second ply below the surface. Only the top four plies were modeled individually to reduce costs, but more detailed models were prepared for later solution.

2. INTERLAMINAR STRESS FINITE ELEMENT MODELING

Recently finite element modeling approaches have been suggested for the determination of interlaminar stresses in those problems that cannot be solved using lamination theory [1,2,3]. All of these involve some form of ply-by-ply modeling and an appeal to St. Venants principle to allow a transition to composite laminate modeling away from the region of interlaminar stress gradients. The issues affecting the accuracy and efficiency of computational models for this class of problems are reviewed in this section and a sysematic approach is presented for use with PATCHES-III. First a variable property simulation of laminate force-deformation behavior is introduced that can represent all 45 elastic constants necessary to characterize general anisotropic laminate deformations [4]. Next a family of linear constraint finite elements are developed from the PATCHES-III tricubic finite element to model the deformations of general laminates. The final and critical development for interlaminar stress gradient analysis is a procedure to transition from discrete 3D ply modeling to 2D+ composite laminate modeling. The features of this new system are then tested using the edge effect demonstration problem and a collection of balanced and unbalanced laminate problems.

2.1 3D COMPOSITE PROPERTIES FOR LAMINATE FORCE-DEFORMATION BEHAVIOR

Pagano [3] has shown that the forces and moments in a general anisotropic laminate are related to laminate strains and curvatures by 45 elastic constants in the equations

$$\sigma_{\mathbf{i}}^{\star} = \overline{C}_{\mathbf{i}\mathbf{j}} \varepsilon_{\mathbf{j}}^{\star} + \overline{B}_{\mathbf{i}\alpha} \kappa_{\alpha}
M_{\beta}/h^{2} = \overline{B}_{\mathbf{j}\beta} \varepsilon_{\mathbf{j}}^{\star} + \overline{D}_{\beta\alpha} \kappa_{\alpha}$$

$$\uparrow \leq \mathbf{i}, \mathbf{j} \leq 6
\text{for}
\alpha, \beta = 1, 2, 4$$
(2.1)

which the PATCHES-III contracted convention is used with ϵ_4 defined as the in-plane shear strain. These equations reduce to the classical plate theory for laminates with monoclinic plies when σ_3 = 0. Once the volume average strains ϵ_i^* and laminate curvatures κ_α have been found, the individual ply stresses can be determined using ply properties C_{ij} and the relations developed by Pagano [4].

The elastic constants in Equation (2.1) are functions of the integrated moments of the ply properties

$$\left[(A_0)_{ij}, (A_1)_{ij}, (A_2)_{ij} \right] = \int_{-h/2}^{h/2} \left[1, z_3, z_3^2 \right] C_{ij}(z_3) dz_3 \qquad (2.2)$$

where the exact expressions for the \bar{c}_{ij} , $\bar{B}_{i\alpha}$, $\bar{D}_{\alpha\beta}$ involve transformations arising from the assumption σ_3 , σ_5 , and σ_6 are each constant through the thickness. The approach developed in the present study uses these same moments to define a variable property composite material in the thickness direction with the same force-deformation behavior. When a finite element of this material is subject to the same volume average strains, ϵ_i , and curvatures κ_α as a laminate modeled using Equation (2.1), the same stress resultants will be produced. The converse of this will not be true in general unless displacement constraints are introduced in the thickness direction to prevent a pseudo edge effect from warping the cross-section. In this study, a linear displacement constraint in the thickness direction is used with the variable property model to define a CCL finite element that predicts the force-deformation behavior of Equation (2.1).

Consider a parametric cubic model for the property variation in the normal direction. This allows up to a sixth degree polynomial in Z_3 when the thickness coordinate function, $Z_3(\xi)$, is also cubic. However, the use of nonuniform geometry models to aid in composite property modeling while feasible [Ref. 5] is not very convenient as it requires solving nonlinear equations. The use of a uniform (i.e., linear) geometry model for $Z_3(\xi)$,

$$Z_3(\xi) = h(\xi - 1/2)$$
 (2.3)

and a parametric cubic for each elastic constant in the thickness direction,

$$C_{ij}^{*}(\xi) = (S_{m})_{ij}^{*} \xi^{4-m} \quad \text{for } 1 \le m \le 4$$
 (2.4)

is adequate for most laminates. In some extreme cases, it may be necessary to use nonuniform geometry modeling to avoid unreasonable values of $C_{ij}^{\star}(\xi)$ at local points through the thickness. However, it is always possible using

uniform models to match the integrated moments in Equation (2.2) which is the essential requirement for finite element modeling. Substituting this property model into Equation (2.2) results in four linear equations

$$(A_n)_{ij} = h^{n+1} \int_0^1 (\xi - 1/2)^n \xi^{4-m} d\xi(S_m)_{ij}$$

for
$$0 \le n \le 3$$

 $1 \le m \le 4$ (2.5)

These can be expressed in matrix form

$$A_{n} = T_{nm} S_{m} \tag{2.6}$$

for any component of $C_{ij}(\xi)$ where

$$T_{nm} = \begin{bmatrix} h/4 & , h/3 & , h/2 & , h \\ 3h^2/40 & , h^2/12 & , h^2/12 & , 0 \\ 7h^3/240 & , h^3/30 & , h^3/24 & , h^3/12 \\ 13h^4/1120 & , h^4/80 & , h^4/80 & , 0 \end{bmatrix}$$
(2.7)

In the special case of a homogeneous laminate, $A_1 = A_3 = 0$ with $A_2 = A_0h^2/12$, the solution of Equation (2.6) results in simply $C_{ij}^*(\xi) = (A_0)_{ij}/h$ as it must. It is possible to simulate A_0, A_1, A_2 with only a quadratic, but the use of the full cubic reduces the number of pathological cases for which $C_{ij}^*(\xi)$ may not be positive definite at all points through the thickness. It is important to note that this is strictly an interpolation problem that can be avoided at the expense of using more interpolation functions.

At this point, there are two issues to be resolved before the approach can be used. First, does the model accurately represent laminate force-deformation behavior, and secondly is there any significant difference with simple constant property models based on the rule of mixtures? The unbalanced laminate analyzed by Pagano [4] for uniaxial loading is used to answer both questions and to illustrate the practical limits of a uniform PC property model.

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Consider the three-ply laminate [60/0/-60] with ply properties

$$C_{11}$$
 = 210.35 GPa C_{22} = C_{33} = 22.30 GPa C_{12} = C_{13} = 7.01 GPa C_{23} = 5.75 GPa C_{44} = C_{55} = 10.34 GPa C_{66} = 4.14 GPa

as taken from Pagano [2.4] where the msi units of that paper are retained for comparison. These ply properties result in the force-deformation properties shown in Table 1 and their inverse shown in Table 2. Under uniaxial load, for example, the laminate twists and stretches with the deformations given by the corresponding column in Table 2. Consider now the representation of this behavior using $C_{ij}^{*}(\xi)$ determined from Equation (2.6). These property functions in PC point format are displayed in Table 3. Note that the extensional moduli were not determined from Equation (2.6), but are the same constants as in Table 1. The reason for this is evident in Figure 1 which shows the PC function for $C_{11}^{\star}(\xi)$. The discrete ply C_{11} properties vary so sharply that their standard deviation, 14.59, is larger than their mean value, 13.66. To represent A_0 , A_1 , and A_2 in this case, a uniform PC function must overshoot zero near the upper and lower surfaces which if used would violate positive definite C_{ii} requirements at these points. To avoid this, only A_0 and A_1 were simulated by the property model in Table 3, and as a result the bending-curvature properties $(M_1, M_2 \text{ loading})$ are not correct. All other force-deformation properties should be exact, and in particular the uniaxial case analyzed by Pagano, σ_1^\star = 1, should be modeled correctly. This was tested using one CCL element in PATCHES-III with Table 3 material properties and loaded by a uniform pressure on opposite faces. The computed laminate strains, ϵ_i^* , and curvatures κ_1 , κ_2 , κ_4 were the same as the Pagano results, Table 4. The results from a unit moment, $M_1 = 1.0$, case are also compared in Table 4, and these deformations are much too low. The bending errors are a direct consequence of the bending stiffness error caused by using constant C_{11} and C_{22} properties from the rule-of-mixtures. In this laminate, [60/0/-60], the stiffness error is over 100 percent, and even in a balanced structural laminate, errors of 10 percent to 20 percent are common.

Consider as an example of this behavior the 48 ply laminate analyzed later in this report. The property distribution (c.f. page 34) shows variable

C14 and C24 coupling through the thickness. This is necessary to account for the bending-shearing coupling $(A_2)_{14}$ and $(A_2)_{24}$ in this laminate which is quite strong. Note that representing this behavior is important to correctly predict the 3D deformation response of the laminate in areas of high local bending.

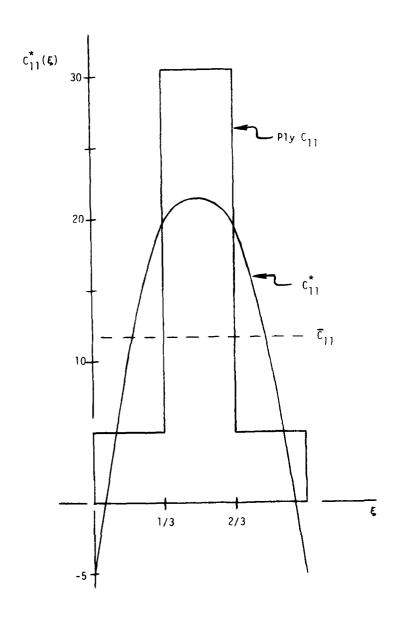


Figure 1. PC Property Modeling Limits

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TABLE 1. THREE-PLY LAMINATE FORCE-DEFORMATION PROPERTIES $(10^6\ PSI)$

| | ε, | ε*2 | ε <mark>*</mark> | ε <mark>*</mark> | ε <mark>*</mark> | ε <mark>*</mark> | к* | ×*2 | κ <mark>*</mark> |
|--------------------------------------|--------|--------|------------------|------------------|------------------|------------------|--------|--------|------------------|
| σ * | 13.656 | 4.232 | 0.925 | | | | | | -0.694 |
| o* 2 | | 13.656 | 0.925 | | | | | | -1.930 |
| * * * 3 | | | 3.234 | | | | | | -0.018 |
| *4 *5 *6 *1 | | | | 4.712 | | | -0.689 | -1.925 | |
| * 5 | | | | | 0.857 | | | | |
| [*] 6 | | (SYM) | | | | 0.857 | | | |
| | | | | | | | 0.494 | 0.450 | |
| 12 | | | | | | | | 1.500 | |
| 1 [*] 2 1 [*] 4 | | | | | | | | | 0.512 |

TABLE 2. THREE-PLY LAMINATE DEFORMATION-FORCE PROPERTIES

| | σ * | σ * | σ * | ♂ * | σ * | σ * | M * | M 2 | M * |
|----------------------------------|------------|------------|------------|------------|------------|------------|-------|-------|-------|
| ε, | 0.082 | 018 | 018 | | | | | | 0.042 |
| ε <mark>*</mark> | | 0.167 | 039 | | | | | | 0.602 |
| | | | 0.325 | | | | | | 0.160 |
| ε*3 ε*4 ε*5 ε*6 *1 | | | | 0.453 | | | 0.141 | 0.439 | |
| ε <mark>*</mark> 5 | | | | | 1.167 | | | | |
| ε <mark>*</mark> | | (SYM) | | | | 1.167 | | | |
| κ * | | | | | | | 2.829 | 668 | |
| ** 2 | | | | | | | | 1.559 | |
| [⋆] 2 [⋆] 4 | | | | | | | | | 4.272 |

TABLE 3. PARAMETRIC CUBIC LAMINATE PROPERTY MODEL*

| | C* _{ij} (0) | C* _{ij} (1/3) | C [*] ij(2/3) | C* _{ij} (1) |
|-----------------|----------------------|------------------------|------------------------|----------------------|
| C ₁₁ | 13.658 | 13.658 | 13.658 | 13.658 |
| c ₁₂ | 4.231 | 4.231 | 4.231 | 4.231 |
| c ₁₃ | .926 | .926 | .926 | .926 |
| C ₁₄ | 3.084 | 1.827 | -1.827 | -3.084 |
| c ₂₂ | 13.658 | 13.658 | 13.658 | 13.658 |
| c ₂₃ | .926 | .926 | .926 | .926 |
| C ₂₄ | 8.581 | 5.085 | -5.085 | -8.581 |
| C ₃₃ | 3.234 | 3.234 | 3.234 | 3.234 |
| C ₃₄ | . 078 | . 046 | 046 | 078 |
| C ₄₄ | 8.284 | 3.523 | 3.523 | 8.284 |
| C ₅₅ | . 550 | 1.217 | 1.217 | .550 |
| C ₅₆ | .385 | .228 | 228 | 385 |
| c ₆₆ | 1.550 | .833 | .833 | 1.550 |
| | | | | |

^{*}Only extensional force-deformation properties simulated.

TABLE 4. FORCE-DEFORMATION COMPARISONS

| | ε* | ε* | ε <mark>*</mark> | ε* | κ ₁ | к ₂ | ^к 4 |
|--------------------------|-------|------|------------------|------|----------------|----------------|----------------|
| Exact $(\sigma_1^* = 1)$ | .0823 | 0184 | 0184 | .0 | .0 | .0 | .0416 |
| $CCL (\sigma_1^* = 1)$ | .0822 | 0184 | 0180 | .0 | .0 | .0 | .0415 |
| Exact $(M_1^* = 1)$ | .0 | .0 | .0 | .141 | 2.829 | 668 | .0 |
| CCL $(M_1^* = 1)$ | .0 | .0 | .0 | .065 | .978 | 193 | .0 |

^{*}Only extensional force-deformation properties simulated.

2.2 CONSTRAINT FINITE ELEMENTS

The CCL finite element used to model laminate force-deformation behavior is one of a family of linear constraint options developed for the PATCHES-III program. The need for low cost modeling in regions of uniaxial or biaxial strain was noted in an earlier report [6] and the new elements in Table 5 provide this capability. They are based on the linear constraints defined in Equation (2.8)

$$P_{i} = T_{i\alpha} P_{\alpha}$$
 $i = 1,2,3,4$ $\alpha = 1,4$ (2.8)

where the coefficients $T_{i\alpha}$ are simply

$$T_{i} = \begin{bmatrix} 1 & 0 & 0 \\ 2/3 & 1/3 & 1/3 \\ 1/3 & 2/3 & 0 & 1 \end{bmatrix}$$
 (2.9)

The same coefficients apply to all three parametric coordinates and, in general,

$$P_{ijk} = T_{i\alpha} T_{i\beta} T_{ky} P_{\alpha\beta\gamma}$$
 (2.10)

If constraints are introduced in only two coordinates,

$$P_{ijk} = T_{i\alpha} T_{j\beta} \delta_{k\ell} P_{\alpha\beta\ell}$$
 (2.11)

and for only one constraint

$$P_{ijk} = T_{i\alpha} \delta_{jk} \delta_{km} P_{\alpha \ell m}$$
 (2.12)

TABLE 5. PATCHES-III FINITE ELEMENT ADDITIONS

| Displacements* | Nodes | Geometry | Properties |
|----------------|-------|----------|------------|
| LLL | 8 | ccc | ccc |
| LLC | 16 | CCC | CCC |
| LCC | 32 | CCC | CCC |
| CCC | 64 | CCC | CCC |

^{*}Any combination of L and C is available; L = Linear, C = Cubic.

Two key issues affecting the development of the new family of elements are how to efficiently generate their stiffness matrices and how to connect them to each other. After some early confusion, it was determined that all linear constraints can be applied before integration with the same result as when they are applied after integration. This greatly reduces the cost of generating their stiffness matrices. It is simple to demonstrate this equivalence in one dimension where obviously

$$K_{\alpha\beta}^{P} = \int T_{i\alpha} F_{i}(\xi) F_{j}(\xi) T_{j\beta} d\xi$$

$$= T_{i\alpha} \int F_{i}(\xi) F_{j}(\xi) d\xi T_{j\beta}$$

$$= T_{i\alpha} K_{i,i}^{P} T_{j,i\beta} \qquad (2.13)$$

but in higher dimensions interpolatory quadrature is used in PATCHES-III and this caused some concern at first.

The second issue was resolved by automating the generation of interface constraints between elements of different dimension. This allows the user of PATCHES-III to specify linear constraints on any element or group of elements by simply placing a mnemonic of the type listed in Table 5 on the connectivity card for that element. The program first generates all explicit mesh point constraints and then on a second pass generates all interface or implicit constraints. This requires extensive checking for conflicts, and in order to reduce their incidence, all three displacement components are constrained alike. At every constrained mesh point, one of the Equations (2.10)-(2.12) is automatically generated and applied to all affected matrices.

2.3 INTERLAMINAR MODELING

The use of substructuring to transition from discrete ply molding to composite laminate modeling was recently investigated by Wang and Crossman [2]. The same technique has been used for other composites under the name "minimechanics" [7] and, of course, it has been used for years in aircraft structural analysis. Substructuring in the case of laminates must account for two transition conditions: one along a plane defined by a ply, and the other along a plane normal to the laminate. The first case requires no transition in the shape of the finite element mesh and appears to work well for edge

effects caused by uniaxial loading [2]. Substructuring for the second case will require mesh changes, as well as the transition to composite properties. An approach using the new CCL finite element with laminate properties is evaluated in this section.

Consider the interlaminar stress problem analyzed in the PATCHES-III User's Manual. First, the accuracy of the new constraint finite elements will be demonstrated for the same four element model using the original properties. The two interior elements were constrained to be all linear (LLL) and the two edge elements were constrained linear in the direction of the load (LCC). No significant change in the earlier results should occur because the displacement solution has the same form in these regions. Normal stress comparisons in Table 6 show this to be true even though the model has been reduced to only 92 degrees-of-freedom. In fact, the assumption of linear displacements in the thickness direction (LCL) also was used with very little mid-surface normal stress error as comparisons in Figure 2 show. Interlaminar stress gradients between the $0^{\rm O}$ and $90^{\rm O}$ plies, however, are drastically changed in the LCL model as the shear stress comparisons in Table 7 show. Note that the interlaminar shear stress appears to contain a singularity at the free-edge between the 00 and 90^{0} plies, σ_{23} (o,b,h). The uniform mesh model gives the mean value of σ_{23} , but one element cannot model this response and return to zero at the freeedge. A nonuniform mesh element does a better job of matching this condition and produces about as much accuracy as can be expected from a single LCC element.

TABLE 6. INTERLAMINAR NORMAL STRESS COMPARISONS*

| ξ/2h** | CCC/CCC | LLL/LCC | LLL/LCL |
|--------------------------------|--|--|---|
| | (428 D.O.F.) | (92 D.O.F.) | (44 D.O.F.) |
| 0 1/3 2/3 1 2 3 | +2.95 26 43 16 02 + .01 | +2.89 27 44 05 03 01 + .02 | +2.76 31 25 03 03 00 02 |

^{*}Comparisons at midsurface between 90° plies. ** $Z_2 = b - \xi$; distance from free-edge.

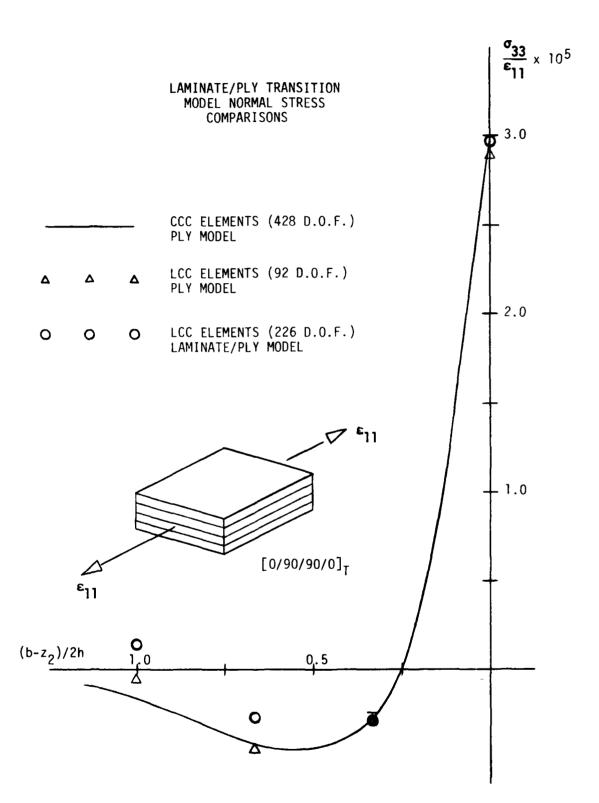


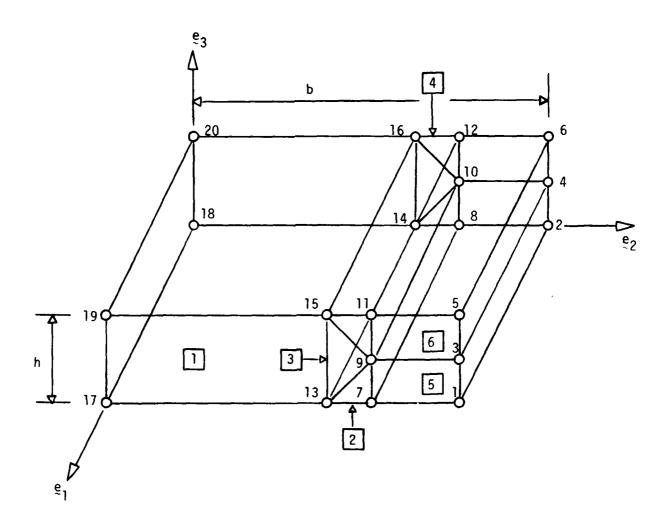
Figure 2. Interlaminar Normal Stress Comparisons

TABLE 7. INTERLAMINAR SHEAR STRESS COMPARISONS*

| ξ/2h | CCC/CCC Uniform | LLL/LCC Uniform | LLL/LCL Uniform | ξ/ 2 h | LLL/LCC Nonuniform** |
|------|--------------------|--------------------|--------------------|---------------|-------------------------|
| 0 | -2.72 | -2.72 | 33 | 0 | + .58 |
| 1/3 | 85 | 85 | 47 | 1/9 | -1.94 |
| 2/3 | 60 | 60 | 42 | 4/9 | 67 |
| i | + .03 | + .03 | +.03 | ì | 31 |
| 2 | 05 | 05 | 04 | 2 | 03 |
| 3 | 02 | 02 | 02 | 3 | 02 |
| 4 | .00 | .00 | .00 | 4 | .00 |

^{*}Comparisons between 90° and 0° plies. **Z $_2$ = 6 + 4 ξ - 2 ξ 2 (edge elements only).

Consider now the transition from discrete ply properties to a composite property model C_{ij}^* away from the free-edge. A six element model shown in Figure 3 was used to determine model accuracy when transitions of this type are made using the present constraint finite elements. Note the two edge elements are exactly as before, and the transition mesh uses three wedge shaped elements. Rule-of-mixture composite properties were used for elements number one and three, and all other elements use local ply properties as in the original model. The interlaminar stresses at the free edge changed very little, Figure 2. At the interface, of course, small local shear stresses occur because of the discrete change in properties. Their maximum value is less than one-half of one percent the applied axial stress. Essentially, the same results were obtained using $C_{i,j}^{\star}(\xi)$ variable composite properties. Interestingly, the two ply laminate properties were easier to model with a cubic than the three ply laminate because the [0/90] properties are an odd function. These results indicate the transition from discrete ply modeling to composite laminate modeling can be made using the new constraint elements.



| ELEMENT DISPLACEMENTS | | PROPERTIES |
|-----------------------|-----|----------------|
| 1 | LCC | LAMINATE , LLC |
| 2 | LCC | PLY , 90° |
| 3 | LCC | LAMINATE , LLC |
| 4 | LCC | PLY , 00 |
| 5 | LCC | PLY , 90° |
| 6 | LCC | PLY , 00 |

Figure 3. PATCHES-III Transition Model for Interlaminar Stress Analysis

3. DISBONDED LAMINATE ANALYSIS

The effects of an unsymmetric disbond on laminate deformations and associated internal load redistributions around the disbond are investigated for a thin laminate. An important aspect of the unsymmetrically disbonded laminate problem <u>not</u> considered in the present investigation is the effect of laminate thickness. There is considerable experimental evidence, Williams [8, 9], that the strength of thick laminates (~50 plies or more) is sensitive to internal disbonds in both unidirectional tape and bidirectional fabric materials of graphite-epoxy. These remarks refer to compressive not tensile loads and to disbonds between plies that occur without fiber damage. Results from the present investigation suggest thin laminates (~10 plies or less) will experience a loss in stiffness caused by out-of-plane bending before any stress critical failure. The laminate in this study is an eight ply graphite-epoxy flat panel with a small circular disbond ($T/R \cong 0.1$) located three plies from the surface and at the center of the panel. A ply-by-ply modeling of this problem is used to study interlaminar force-deformation behavior in the vicinity of the disbond with CCL finite elements. No attempt was made to include in the model any strain singularities that might exist, although this could have been done had fracture mechanics been the focus of the study. The objective was to determine the deformations and internal load redistribution caused by an unsymmetric disbond in a compressively loaded thin laminate.

The disbonded laminate problem was analyzed early in the study using symmetry boundary conditions on two planes. Prototype models indicated a definite skewed displacement response would occur and that the use of symmetry planes would inhibit some response modes. Unfortunately, PATCHES-III was limited to 50 elements at that time and no practical means of avoiding symmetry modeling was available. The resulting symmetry model solution is essentially a 3D laminate solution with interlaminar stress gradients distorted at the origin by the symmetry boundary conditions. However, the force-deformation behavior of the laminate is represented reasonably well and several interesting conclusions can be drawn from the symmetry solution.

3.1 PROTOTYPE MODELS

A schematic of the idealized disbonded laminate analyzed in this study is shown in Figure 4. The disbonded region is at the center of a square

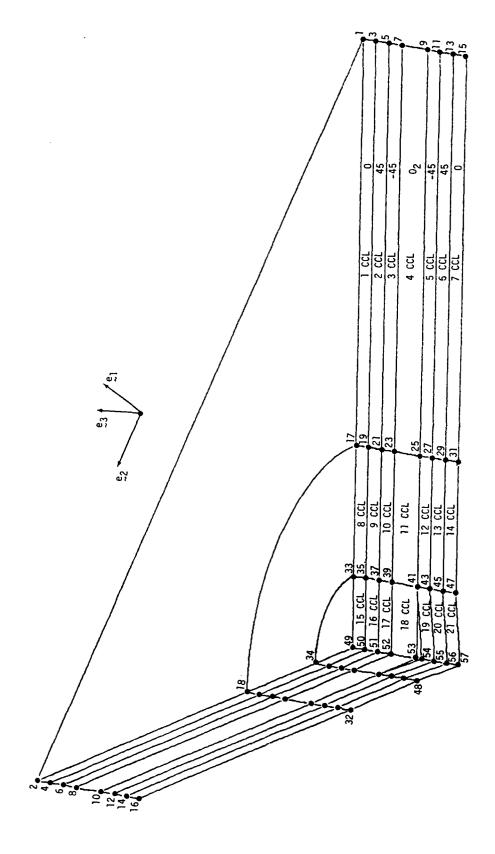


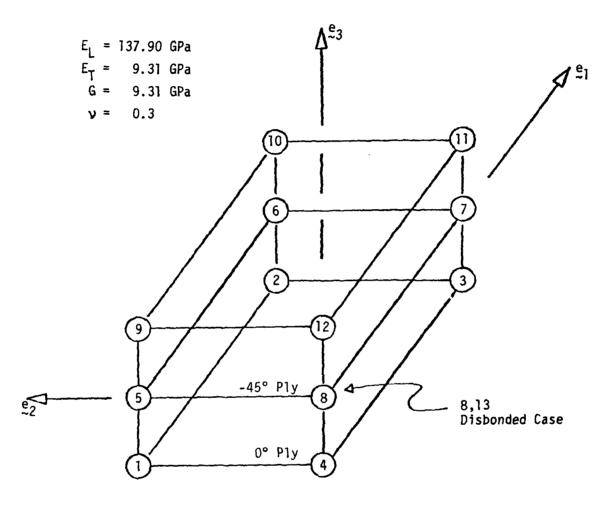
Figure 4. Disbonded Laminate Schematic Using Diagonal Symmetry

laminate between plies five (-45°) and six (0°) . The region is circular with a bubble shape whose profile is given by

$$Z_3 = t_p(1+\cos r\pi/R)/2$$
 (3.1)

where t_p is a ply thickness and the maximum radius is R = 1.27 cm (0.5 in.). Note that this shape is tangent to the adjacent bonded region and corresponds to an elastically warped surface. A catenary shape would correspond to inelastic fiber damage at the boundary of the disbonded region. The laminate is 15.24 cm (6 in.) square and uniformly compressed 0.01524 cm (.006 in.) in both the e_1 and e_2 directions. At the loaded edges, the normal displacement component, u_3 , is restrained to zero. The disbonded plies [-45/45/0] are unbalanced and after their separation, the remaining plies $[0/45/-45/0_2]$ are also unbalanced. As in the unbalanced three-ply laminate analyzed by Pagano, there will be stretching-twisting coupling that must be considered in selecting a symmetry model. It is also important to examine the inplane force changes caused by a disbond to assist in preparation of the ply-by-ply model of the laminate.

A small two element model was used to examine interply force changes when a -45° ply and a 0° ply disbond locally. The prototype model, Figure 5, consists of a -45° ply and a 0° ply in one quadrant of the laminate (7.62 cm) loaded by a uniform strain ε_{11} = 0.001. To simulate the disbond, the two elements were disconnected at node 8 corresponding to the center of the laminate. The two elements were type LLL and the nodal forces at the interply corners were used to measure the effect of the disbond. These are shown in Figure 6 and indicate only a small constraint force is relaxed by a disbond at node 8. Note, however, that the effect is to increase the transverse load in the 0° ply which carries most of the load associated with the applied ϵ_{11} strain. It is obvious that a larger change will occur if either nodes 5 or 7 are cut instead of node 8 which is analogous to a disbond at node 8 in a +45/0° layup. Results from this case, Figure 7, do show a larger change, but it reduces the transverse load in the 0° ply. The directional dependence in stiffness produces a directional dependence in the effect of a disbond under any given load. In this case of a circular disbond, Figure 4, under biaxial load, it appears that interply constraint forces will vary around the perimeter with relative maximums or minimums every 45°.



 $u \cdot e_1 = 0$ on Faces 4-8-5-1, 13-12-9-5 $u \cdot e_1 = 0.03$ on Faces 3-7-6-2, 7-11-10-6

Figure 5. Two-Ply Disbond Prototype Model for Forces

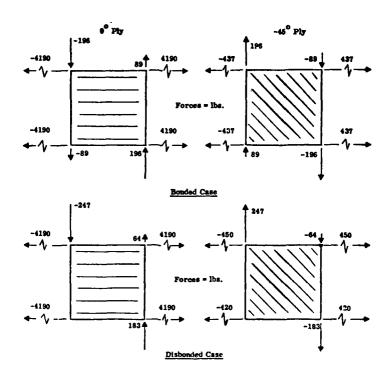


Figure 6. Interply Forces: [0/-45] Uniaxial Load

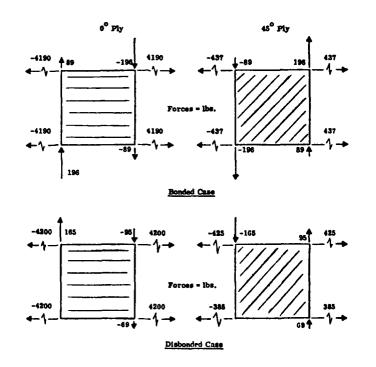


Figure 7. Interply Forces: [0/45] Uniaxial Load

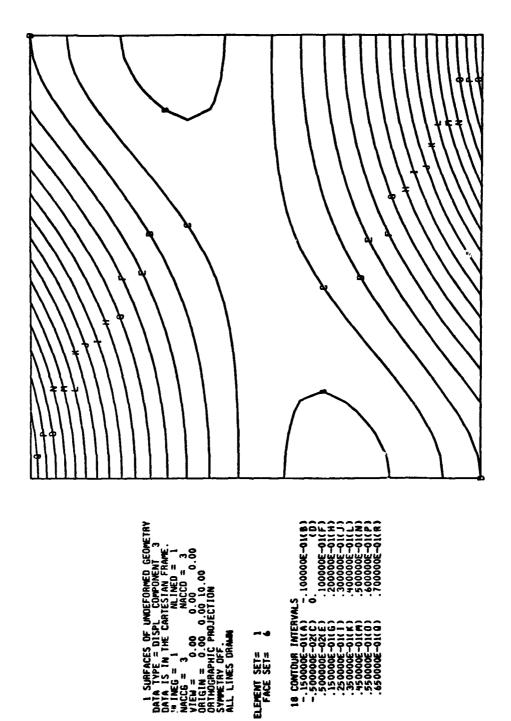
Next, a three-ply [0/45/-45] laminate model was used to examine out-of-plane warping under inplane loading of the unbalanced disbond material. Each ply was modeled with CCL elements, and the dimension of the square model was made equal the radius of the disbond. Under uniaxial load there is elastic coupling between stretching and twisting that caused large out-of-plane displacements because of the low bending stiffness of a three-ply laminate (the plies are only 0.01778 cm thick). The stresses, strains and deformations of this [0/45/-45] laminate are all either asymmetric or symmetric about axes at $\pm 45^{\circ}$ to axes e_1 , e_2 , Figure 8. These results and the [0/45] constraint force results indicate diagonal symmetry rather than Cartesian symmetry should be used in modeling force-deformation behavior of the disbonded laminate.

Another observation from the unbalanced prototype models was the extremely slow convergence of the conjugate-gradient solution procedure. The membrane response accounts for 99.7 percent of the potential energy and this result is converged in less than N cycles. However, the low energy out-of-plane bending response required almost 4N cycles to converge with little change to principal stresses or strains. Note that the interply normal forces are two orders of magnitude smaller than the inplane forces. The models were changed to CCC elements to check for purely numerical ill-conditioning and the same slow convergence was observed. No solution to the pathological computational behavior caused by weak out-of-plane coupling was found.

3.2 LAMINATE FINITE ELEMENT MODELS

The dimensions of the laminate were specified to insure no coupling between the displacement boundary conditions and deformations in the vicinity of the disbond. The laminate is square of width 15.24 cm (6 inches) with a disbond radius of 1.27 cm (.5-inch) between plies five and six, Figure 4. The plies are graphite-epoxy of thickness 0.01778 cm (0.007-inch) with their nominal elastic properties as given in Table 8. The stacking sequence for the laminate is $[0,45,-45,0_2,-45,45,0]_T$ with the disbond between a 0° ply and a -45° ply. The edges are loaded by imposing a uniform inplane displacement of 0.00762 cm (0.003-inch) which produces biaxial compression.

A control model of the laminate was constructed using the same mesh in the Z_1, Z_2 plane as in Figure 4, but with only four plies $[0,45,-45,0]_S$ through the half thickness. The purpose of this model was to provide beforeand-after data for comparison with the disbonded results, in particular the



UNDALANCED LANIMATE

Unbalanced Laminate Bending Displacements Figure 8.

ELEMENT SET= FACE SET=

TABLE 8. GRAPHITE-EPOXY NOMINAL ELASTIC PROPERTIES

| E ₁₁ = 137.90 GPa (20 msi) | ν ₁₂ = 0.3 | G ₁₂ = 4.83 GPa (.7 msi) |
|---|-----------------------|---|
| $E_{22} = 9.31 \text{ GPa } (1.35 \text{ msi})$ | $v_{13} = 0.3$ | $G_{13} = 4.83 \text{ GPa } (.7 \text{ msi})$ |
| $E_{33} = 9.31 \text{ GPa } (1.35 \text{ msi})$ | $v_{23} = 0.3$ | $G_{23} = 4.83 \text{ GPa } (.7 \text{ msi})$ |

internal forces. At the same time the control model shows the magnitude of the interlaminar constraint forces caused by the use of diagonal symmetry, Table 9. The forces F_2 should be exactly zero if the interlaminar deformations are zero at the center of the laminate. Instead, a large self-equilibrating pair of forces occur between the 0° plies and the $\pm 45^\circ$ plies. These forces distort the interlaminar stresses at the origin. It should be noted that F_2 sums to zero between the $\pm 45^\circ$ and the use of diagonal symmetry for these plies is consistent with material symmetry axes.

TABLE 9. BONDED LAMINATE CONSTRAINT FORCES THROUGH THE THICKNESS AT r = 0

| | $Z_3 = 4t_p$ | $Z_3 = 3t_p$ | $Z_3 = 2t_p$ | $z_3 = t_p$ | $Z_3 = O_p$ |
|----------------|--------------|--------------|--------------|-------------|-------------|
| NODE | 49 | 50 | 51 | 52 | 53 |
| Fl | 54.98 N | 100.44 N | 102.00 N | 100.22 N | 50.18 N |
| F ₂ | 0. | 43.41 N | 0. | -43.41 N | 0. |

The control model also demonstrates that even the small biaxial strain ($\epsilon \cong 0.1\%$) imposed at the edges of the panel is sufficient to buckle the laminate elastically. An engineering check was made using isotropic simply supported plate formulas with the stiff direction properties. This calculation indicates the laminate will buckle at $P_1 = P_2 \cong 2000$ N, while the applied load is P > 10,000 N. The exact solution for buckling of a laminated anisotropic plate [10] would be lower because the small number of plies accentuates the bending-stretching coupling. It seems likely that the disbond will reduce the buckling load even further or possibly cause a large deflection stability problem in a laminate this thin, T = 0.142 cm (0.056-inch).

3.3 DISBONDED LAMINATE RESULTS

The effect of the disbond on the interlaminar force distribution is quite small and the effect on bending deformation is quite large. The centerpoint initial displacement caused by the disbond is $0.5 t_n$ before any load is applied. The centerpoint elastic displacement after loading to a biaxial strain of only 0.1 percent is an additional 0.4 $t_{\rm p}$, Figure 9. In other words, a unit imposed displacement (biaxial) in the plane of the laminate produces almost a unit normal displacement at the center of the laminate. This result suggests the disbond will initially lead to a stiffness critical failure rather than a stress failure. The reason for this behavior seems to be the extremely low bending stiffness which produces large out-of-plane deformation when the disbond couples the membrane and bending displacement response. It is important to note that the potential energy change between the bonded and disbonded cases is less that one milli Joule, Table 10, which corresponds to an energy release rate of less than 1 N/m. In contrast, the critical energy release rate for an edge delamination, Reference 11, is will over 100 N/m for an 11 ply graphite-epoxy laminate.

TABLE 10. DISBONDED LAMINATE ENERGY CHANGE

| | | Potentia | l Energy | |
|-----------|-------------|----------|-----------|--------|
| Bonded | 3.958455 Jo | oules | (35.03528 | inlbs) |
| Disbonded | 3.958069 Jo | oules | (35.03186 | inlbs) |

The small redistribution of internal forces is evident in Table 11 which shows the center point constraint forces through the thickness. Note that the forces at nodes 53 and 54 are summed when the disbond is closed. Note also that the force at node 53 in the bonded model must be added to node 52 for comparison with the disbond results. This is because only one element was used for the two center plies, Figure 4, in the disbond model. A check of the force changes through the thickness was also made at r = R, Table 12.

To a large extent the symmetry boundary conditions inhibit in-plane redistribution and to some extent interlaminar normal force changes. Considering the large normal displacements, Figure 9, one might expect an

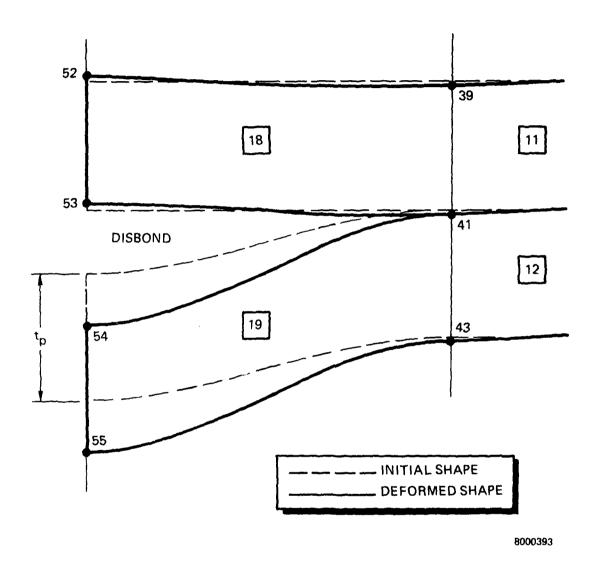


Figure 9. Normal Deformations in the Plies Adjacent to the Disbond

TABLE 11. DISBONDED LAMINATE CONSTRAINT FORCE CHANGES THROUGH THE THICKNESS AT r = 0

| | | Bonded | | | Disbonded | | |
|------|-----------------|----------------|----------------|---|---------------------------|----------------|----------------|
| Node | z ₃ | F ₁ | F ₂ | | ^Z ₃ | F ₁ | F ₂ |
| 49 | 4t _p | 54.98 N | 0. | N | 4t _p | 44.54 N | 1.01 N |
| 50 | 3t _p | 100.44 N | 43.41 | N | 3t _p | 103.68 N | 45.75 N |
| 51 | 2t _p | 102.00 N | 0. | N | 2t _p | 105.69 N | 22 N |
| 52 | tp | 100.22 N | -43.41 | N | t _p | 160.78 N | -46.58 N |
| 53* | 0 | 50.18 N | 0. | N | -t _p | 112.41 N | 90 N |
| 54 | | | | | -1.5 t _p | 43.62 N | -41.33 N |
| 55 | | (Symmo | etric) | | -2.5 t _p | 90.67 N | 3.22 N |
| 56 | | | | | -3.5 t _p | 89.49 N | 38.57 N |
| 57 | | | | | -4.5 t _p | 42.46 | 1.43 N |

*Node 53 is at $Z_3 = 0$ in the bonded laminate case.

interlaminar normal force to occur at r = R. As the data in Table 12 show, only a small increase in the small interlaminar normal force occurred at node 41. A possible explanation is that the symmetry boundary conditions at the center prevents nodes 53 and 54 from moving relative to each other in-plane as they probably would in a complete solution. If such deformations occur, they would create transverse shear gradients in-plane and equilibrium would require a normal stress gradient in the thickness direction,

$$\sigma_{31,1} + \sigma_{32,2} + \sigma_{33,3} = 0$$
 (3.2)

The magnitude of the spurious F_2 force at node 54 suggests this deformation could be significant and would certainly increase the energy release rate. However, it would have to increase by a factor of 100 before the delamination would propagate if the data in Reference 11 apply. This uncertainty can only be resolved by analyzing a complete model without the assumption of symmetry conditions.

TABLE 12. DISBONDED LAMINATE CONSTRAINT FORCE CHANGES THROUGH THE THICKNESS AT r = R

| | | Bonded | | | Disbonded | |
|------|------------------|---------------------------------|------------------|------------------|---------------------------------|----------------|
| Node | z ₃ | F ₁ = F ₂ | F ₃ * | z ₃ | F ₁ = F ₂ | F ₃ |
| 33 | 4t _p | 12.18 N | 0. N | 4t _p | 14.01 N | 0. N |
| 35 | 3t _p | 74.54 N | ±0.56 N | 3t _p | 75.17 N | ±0.07 N |
| 37 | 2t _p | 68.29 N | ±1.12 N | 2t _p | 69.10 N | ±0.14 N |
| 39 | ^t p | 33.09 N | ±0.84 N | t _p | 33.13 N | ±0.93 N |
| 41 | -t _p | | | -t _p | 32.16 N | ±1.49 N |
| 43 | -2t _p | (Symme | tric) | -2t _p | 67.28 N | ±0.84 N |
| 45 | -3t _p | | | -3t _p | 73.04 N | ±0.34 N |
| 47 | -4t _p | | | -4t _p | 12.19 N | 0. N |

 $^{{}^{\}star}\mathsf{F}_3$ forces are self-equilibrating interlaminar forces.

The interlaminar stress results in keeping with the force results show only small changes in their distribution. On a percentage basis, the largest change occurs in the bottom ply where the small σ_{LT} shear stress is doubled. A plot of this stress component around the disbond at r=R, Figure 10, shows relative maximums and minimums at $\pm 45^{\circ}$ as expected. The use of symmetry boundary conditions also caused local distortions at the origin in the stress results that make their magnitudes questionable. However, these conditions affect the bonded and disbonded cases equally.

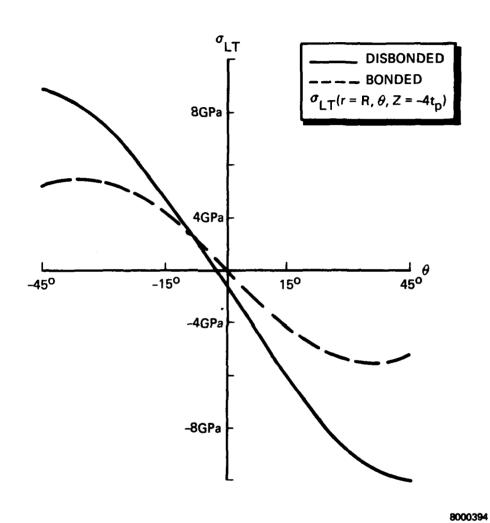


Figure 10. Bottom Ply Shear Stress Changes

4. LAMINATE IMPACT ANALYSES

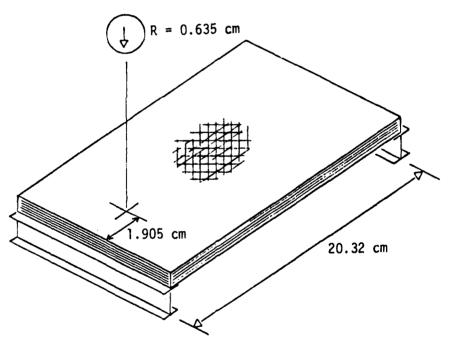
At low velocities impact damage can be modeled as a quasi-static response problem using Hertzian contact pressures in most cases. This approach is used in the present study, and Greszczuk, Reference 12, gives an excellent account of the method and its rationale. The only analytic results available are for the elastic half space problem which was recently solved for a transversely isotropic material, Reference 13. One effect of transverse isotropy shown graphically in that paper is a shift of the maximum Von Mises stress from the impact centerline to a radius approaching the contact radius. The significance of this stress parameter for a composite is dubious, but it does indicate a trend of increasing transverse shear below the contact radius, which also occurs in the present results.

The disbonded laminate analysis problems caused by symmetry displacement boundary conditions led to the use of a full 360 degree finite element model for the impact problem. Another factor in this decision was the earlier PATCHES-III sandwich panel impact analysis [6] which also showed spurious constraint forces between the 0° and 45° plies on the centerline of a double symmetry model. In order to model a full 360 degrees, ply-by-ply, the program was modified to allow several hundred solid elements. Unfortunately, the computer budget available was sufficient to analyze only a 32 element model after all computational problems were resolved. These results, however, provide the first in-depth picture of the asymmetric interlaminar stress gradients that occur in a thick laminate impacted off-center. More detailed models were prepared for future analysis and these are in an appendix to this report.

4.1 THICK LAMINATE COMPOSITE RESPONSE

A 48 ply graphite-epoxy laminate supported between two rigid spars is shown schematically in Figure 11 being impacted by a steel sphere. The impact point is close to a spar and in this problem a contact force of 4448 N was specified. Considering the large number of plies in this problem, not all plies can be modeled for any given analysis. This practical consideration becomes even more critical when symmetry conditions cannot be used. First a variable property 3D laminate model was used to determine the bending deformations in the vicinity of the impact site. This analysis used a large but finite width strip to avoid boundary condition effects, Figure 12. The 3D

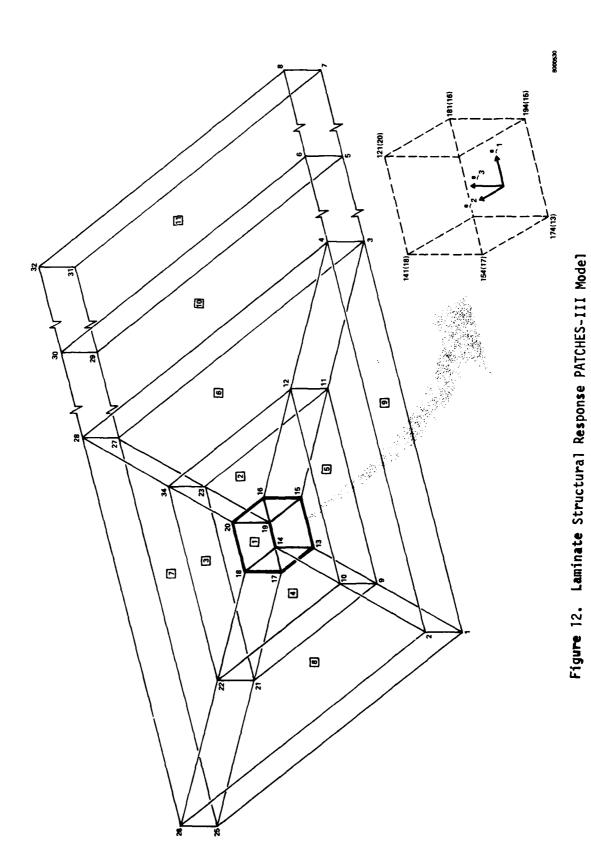
laminate analysis produced large transverse shear stresses at the impact site approaching the bending stresses in magnitude. Laminate force-deformation properties were modeled using the $C_{ij}^{\star}(\xi)$ formulation derived earlier, Table 13. As these data indicate, only the off-diagonal coupling terms vary appreciably through the thickness.



LAMINATE: [±45/0₂/±45/0₂/±45/0/90]_{2s}

GRAPHITE-EPOXY

Figure 11. Thick Laminate Impact Schematic



-33-

TABLE 13. PARAMETRIC CUBIC THICK LAMINATE PROPERTY MODEL*

| | $C_{ij}^{*}(0) = C_{ij}^{*}(1)$ | $C_{ij}^{*}(1/3) = C_{ij}^{*}(2/3)$ |
|-----|---------------------------------|-------------------------------------|
| C11 | 72.622 | 71.650 |
| C12 | 21.712 | 14.906 |
| C13 | 2.999 | 2.910 |
| C14 | 2.427 | -0.807 |
| C22 | 24.173 | 36.887 |
| C23 | 2.379 | 2.468 |
| C24 | 2.427 | -0.807 |
| C33 | 9.432 | 9.432 |
| C44 | 22.091 | 16.279 |
| C55 | 3.771 | 3.771 |
| C66 | 3.744 | 3.744 |
| C66 | 3.744 | 3.744 |

^{*}Stiffness in GPa units (1 Psi = 6894.757 Pascals)

4.2 IMPACT SITE MODEL

The composite laminate displacement response for element number one, Figure 12, provided boundary conditions for the impact site model shown in Figure 13. The contact radius was determined using both the Greszczuk [12] equations and the Dahan and Zarka [13] equations. The latter have typographical errors that were reconciled using the limiting case of an isotropic half-space to yield

$$r_0 = \left[\frac{3FR}{2} \left(\frac{\delta_1 - \delta_2}{2} + \frac{1 - v^2}{E} \right) \right]^{1/3}$$
 (4.1)

where δ_1 and δ_2 are lengthy algebraic functions of the half-space flexibilities S_{ij}^* . Substituting the laminate S_{ij}^* in these expressions gave

$$\delta_1 = -8.06 \times 10^{-12} \text{ m}^2/\text{N}$$

$$\delta_2 = -68.09 \times 10^{-12} \text{ m}^2/\text{N}$$

The remaining terms in Equation (4.1) are the force F = 4448 N, the sphere

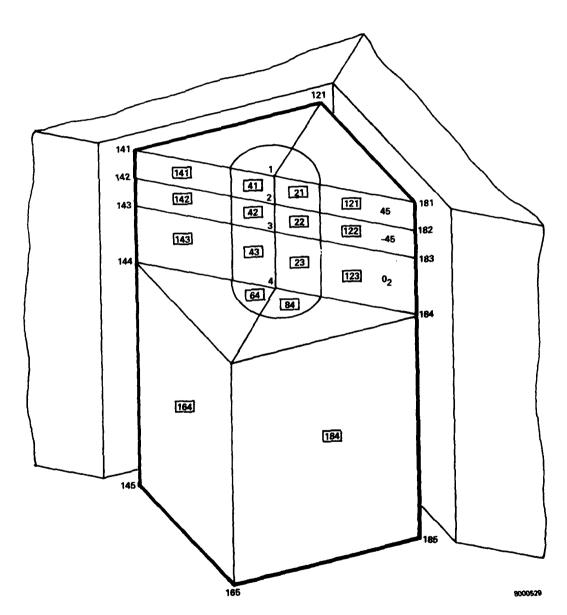


Figure 13. Impact Site PATCHES-III Medel

radius R = 6.35 mm and the elastic constants E = 206.84 Gpa and ν = 0.3 for the sphere. These data produced a contact radius of r = 1.134 mm. The Greszczuk equation for contact radius

$$r_0 = \left[\frac{3\pi}{4} FR \left(k_1 + k_2\right)\right]^{1/3}$$
 (4.2)

also uses the $\mathbf{S_{i\,j}^{\star}}$ to compute related coefficients $\mathbf{k_{1}}$ and $\mathbf{k_{2}}$ which are

$$k_1 = 1.4004 \times 10^{-12} \text{ m}^2/\text{N}$$

 $k_2 = 19.8173 \times 10^{-12} \text{ m}^2/\text{N}$

producing a contact radius of 1.122 mm. The smaller radius was used to model the impact site so that $T/R\cong 5.65$.

Several disastrous attempts at analyzing the impact site model, Figure 13, were made using CCC elements for the upper plies and CCL elements for the subsurface laminate model. These attempts never quite converged and the deformations were highly distorted. After considerable effort it was determined that the model had pathological stiffness discontinuities in the thickness direction that were easily removed once found. At the impact site the elements through the thickness must be either all CCC or all CCL to avoid this condition. When the two element types are mixed under impact loading conditions, the CCC elements deform as if they were pressed against a rigid boundary. After resolving this modeling dilemma, excellent results were obtained using all CCC elements.

4.3 INTERLAMINAR IMPACT STRESS RESULTS

It is important to keep in mind that the present results are for a $\frac{small}{s}$ sphere impacting a $\frac{thick}{s}$ laminate, $\frac{t}{R} \cong 5.65$, with sufficient energy to generate a contact force of 4448 N (1000 lbs). The resulting stress-strain gradients are quite high in the upper ply group, Figure 14, especially in material coordinates. Note in particular that the second ply, a - 45, has larger fiber stress gradients in the thickness direction than the top ply, although the magnitudes are slightly lower. The centerline Cartesian shear strains, Figure 15, show rather dramatically that the use of two symmetry planes would have produced large shear errors along the centerline. Interesting, the chordwise shear strain, ε_{23} , is very small indicating one plane of symmetry

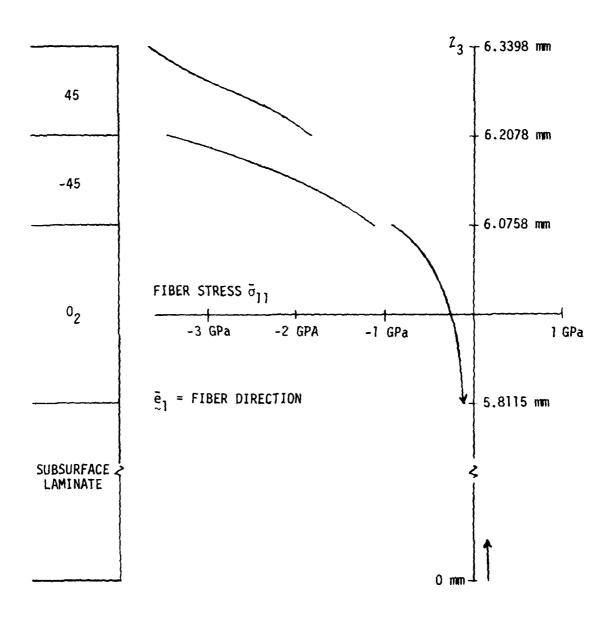


Figure 14. Centerline Fiber Stress

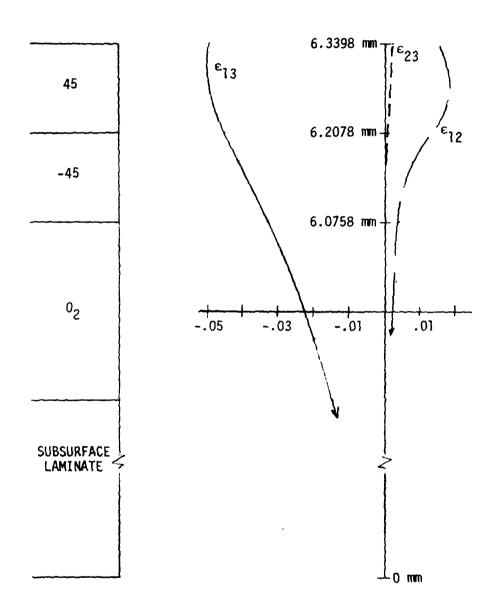


Figure 15. Cartesian Centerline Shear Strains

in the response. If the loading were not in the transverse direction, or if the laminate were unbalanced, even this symmetry condition would not exist.

Consider next the stress distribution around the perimeter of the impact site. The fiber stresses at the top surface as a function of circumferential position, $\bar{\sigma}_{11}(\theta)$, are shown in Figure 16. The tension stresses are confined to a thin layer at the free surface and are probably not a good indicator of actual fiber stresses. When the fiber stresses are averaged through a single ply thickness, Figure 16, the values are all compressive. In either case, the distribution is periodic in θ with a period equal π . The distribution of transverse shear stress in material coordinates is also periodic, Figure 17, but with period equal 2π . The two shear stress components in this figure, $\bar{\sigma}_{13}$ and $\bar{\sigma}_{23}$, were averaged over one ply thickness, the second ply. This ply has the maximum transverse shear in the laminate and it occurs at the impact perimeter not the centerline. Similar behavior was observed by Dahan and Zarka in zinc and cadmium during elastic contact with a steel sphere. These materials also had their low stiffness in the direction normal to the impact surface, but they are not as anisotropic as the laminate material. It is also interesting to note that the two shear stress components $\bar{\sigma}_{13}$ and $\bar{\sigma}_{23}$ are phase shifted by 90° in the circumferential direction. The in-plane ply shear stress $\bar{\sigma}_{12}$ in Figure 18 shows a quasi-periodic distribution of period equal π that is similar to the fiber stress distribution. However, the structural response of the laminate appears to be superimposed on the elastic halfspace response. This would explain the $\bar{\sigma}_{12}$ peak nearest the center of the laminate being slightly smaller. There is also a subharmonic component that may be caused by the composite solution boundary conditions applied to the impact site model. To explore this second order effect would require additional analyses. A summary of the ply material stress maximums is provided in Table 14.

It is interesting to observe the similarities and differences in a 3D laminate solution and a 3D elasticity solution in the impact area. The deformations at the top surface are compared in Figure 19 and show very similar shapes. The maximum 3D laminate displacement is about 85 percent of the elasticity result; however, the laminate strains are an order of magnitude too low at the top surface. The elasticity solution using laminate properties was much closer but still low. The normal strain in this case is quite close, but the in-plane strains are one-half to one-third the ply strain results.

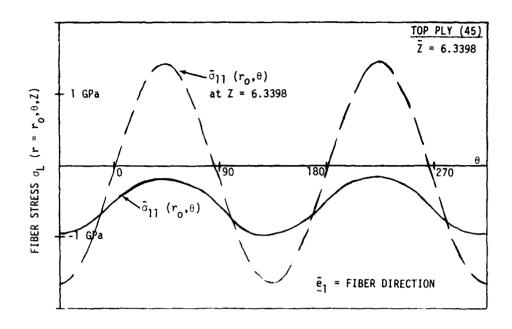


Figure 16. Fiber Stresses Around Impact Perimeter

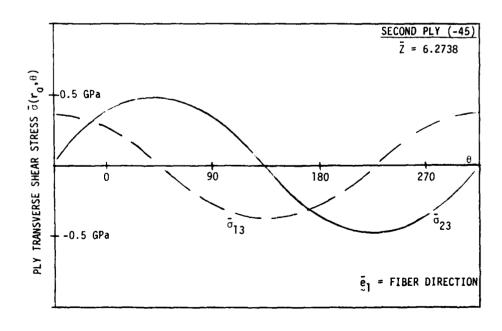


Figure 17. Transverse Shear Stresses Around Impact Perimeter

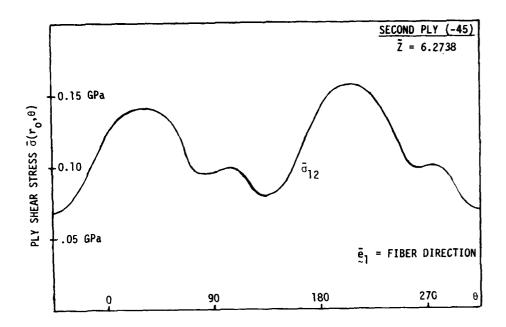


Figure 18. Ply Shear Stress Around Impact Perimeter

TABLE 14. THICK LAMINATE PLY STRESS SUMMARY

| | σμ | αT | ^σ LT | σ _{TT} |
|------------------|------|-----|-----------------|-----------------|
| σ/P _o | 2.05 | .99 | .28 | 28 |
| Node | 1 | 1 | 22,62 | 22,62 |

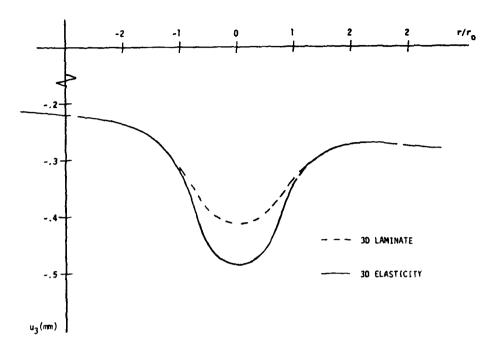


Figure 19. Thick Laminate Local Bending Deformation Comparisons

5. CONCLUSIONS AND RECOMMENDATIONS

A systematic approach to modeling complex interlaminar stress gradients has been developed and applied to two graphite-epoxy laminates. The approach uses variable property 3D finite elements that correctly model force-deformation behavior and constant property 3D finite elements that accurately model ply shear strains. Analyses for this class of problems are difficult, but considerable progress has been made toward solving the computational and modeling issues encountered.

- To model composite laminate force-deformation behavior with 3D finite elements requires, in general, properties that vary through the laminate thickness.
- Interlaminar deformations are <u>not</u> symmetric in a balanced symmetric laminate under symmetric in-plane loads.
- Interlaminar deformations have <u>one</u> plane of symmetry in a balanced symmetric laminate under normal Hertzian contact pressure.

Analysis of the disbonded laminate force deformation behavior indicates a large out-of-plane displacement response occurs without singificant internal force redistribution. Comparison with a bonded laminate control model also indicates very little energy is released by the disbond. The new CCL finite element worked well in this application and should be an effective new tool for analyzing interlaminar force-deformation behavior in the vicinity of defects. It is interesting to note that more accurate results can be obtained on reanalysis by simply changing the element specification to CCC as in the P-Version of the finite element method [14].

Analysis of the thick laminate impacted off-center by a small sphere indicates the maximum transverse shear occurs below the surface in the second ply for the $[\pm 45/0_2/\pm 45/0_2/\pm 45/0/90]_{2S}$ layup. This maximum occurs at the contact radius and both transverse shears in material coordinates have a sinusoidal variation around the perimeter. The fiber stresses also vary sinusoidally, but with one-half the period of the transverse shears. Only the inplane shear stress shows a noticeable effect of the impact site being off-center. The edge closest the spar boundary has slightly higher ply shear. This also occurs in the ply shear strains with the relative increase being

on the order of 10 percent. These results were obtained after several analysis failures caused by mixing CCL and CCC elements in the thickness direction. Based on these results and those from the disbonded laminate the following additional analyses are recommended:

- Analyze the disbonded laminate without any symmetry assumptions.
- Analyze the disbonded laminate with twice as many plies, $[0/\pm 45/0_2/\pm 45/0]_{2S}$, with the disbond in the same location.
- Analyze the impact site with the focus on the bottom ply group.

Now that accurate modeling techniques for finite element analysis of interlaminar behavior have been developed, a combined test/analysis study of the effects of defects is recommended. It would be possible to evaluate energy release rate and possibly other parameters as a measure of how large a delamination can become before unstable propagation takes place. Pre-test analyses could be used to define the critical load conditions and to help locate instrumentation. In order to facilitate this work it would be desirable to interface the PATCHES-III program with a composite material synthesis program [15], [16] for pre- and postprocessing of the finite element results. This would allow basic ply properties and layup data as input and automate the recovery of ply stress-strain data from composite stress-strain results. It would also allow Tsai-Wu, Hill and other failure criteria to be applied to the PATCHES-III results.

6. REFERENCES

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APPENDIX

The PATCHES-III input data for the two laminates analyzed in this report and data modifications necessary to analyze three additional impact cases, Table A-1, are provided in this appendix. The structural response model that provides displacement boundary conditions for the impact site model is included for each case. Also, a schematic of a more detailed model of the impact site that includes the entire 12 plies in the basic repeating ply group is provided.

TABLE A-1. PATCHES-III LAMINATE IMPACT CASES

| Case | Impact Site | Sphere Radius | Contact Radius | Status |
|------|----------------|------------------|-------------------|----------|
| 1 | Z1 = 1.91 cm | 0.635 cm | 1.12 mm | Analyzed |
| 2 | Z1 = 1.91 cm | 2.540 cm | 1.78 mm | Modeled |
| 3 | Z1 = 10.16 cm | 0.635 cm | 1.12 mm | Modeled |
| 4 | Z1 = 10.16 cm | 2.540 cm | 1.78 mm | Modeled |
| | | | | |

PATCHES-III INPUT DATA
DISBONDED LAMINATE MODEL

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PATCHES-III INPUT DATA LAMINATE IMPACT STRUCTURAL RESPONSE MODEL

Case 1. (See Computer Listing)

Case 2. Change gridpoints 9, 11, 13, 14, 17, 19, 21, 23.

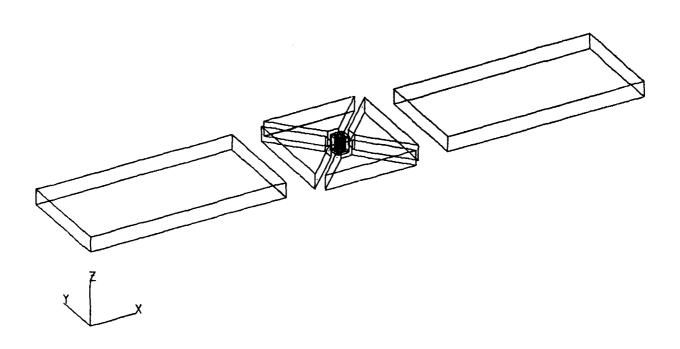
GRID, 9, ,-0.356-2,-0.356-2 GRID, 11, 0.356-2,-0.356-2 GRID, 13, ,-0.178-2,-0.178-2 GRID, 15, 0.178-2,-0.178-2 GRID, 17, ,-0.178-2, 0.178-2 GRID, 19, 0.178-2, 0.178-2 GRID, 21, ,-0.356-2, 0.356-2 GRID, 23, 0.356-2, 0.356-2

Case 3. Change gridpoints 5, 7, 29, 31 and redefine element number 11 data.

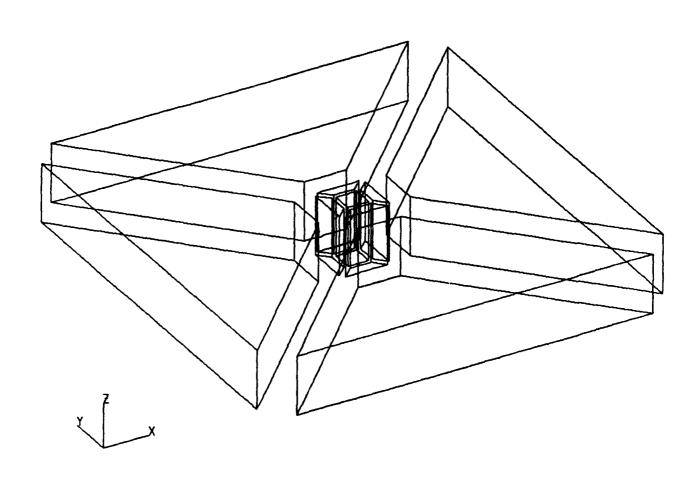
GRID, 5, , 10.16-2,-2.0-2 GRID, 7, ,-10.16-2,-2.0-2 GRID,29, , 10.16-2, 2.0-2 GRID,31, ,-10.16-2, 2.0-2 PATCHQ,11, 7, 31,25, 1 CPDE3 ,11, 7, 31,25, 1 +C11 ,CCL, 8, 32,26, 2 SDC10 , 10,11,123, 7, 8,32,31 SDC10 , 10,11, 2, 7, 1, 2, 8 SDC10 , 10,11, 2, 7, 1, 2, 8 SDC10 , 10,11, 2,31,25,26,32 SDC10 , 10,10,123, 5, 6,30,29 (delete) SDC10 , 10, 8,123, 1, 2,26,25

Case 4. Combine the changes for Cases 2 and 3.

LAMINATE STRUCTURAL RESPONSE MODEL FOR CASES 3 AND 4



IMPACT SITE STRUCTURAL RESPONSE MODEL FOR ALL CASES



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CASE CONTROL DATA

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SHRITTLE & 48 PLY MODEL & VARTABLE PROPERTIES
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| 30. | GPID | 23 | | 2 | | 2-7 | | |
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BULK DATA CAROS CONTRACTOR CONTRA

FTELN-1,FTELD-2,FIELD-3,FIELD-4,FIELD-5,FIFLD-6,FIELE-7,FIELD-8,FTELD-9,FIELD-10

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PATCHES-III INPUT DATA LAMINATE IMPACT IMPACT SITE MODEL

- Case 1. (See Computer Listing)
- Case 2. (See Computer Listing Page A-29 for gridpoint changes).
- Case 3. Same as Case 1 with new structural response input data.
- Case 4. Same as Case 2 with new structural response input data.

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| 253* | | 241 | b √1 | 122 | 121 | 141 | 142 | | | |
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| 270- | | ā | س | 145 | 144 | 164 | 165 | | | |
| 275- | - | 91 | | 162 | 161 | 181 | 182 | • | | |
| 276- | DPATO | 191 | ~ | 162 | 161 | 181 | 182 | 7 | | |
| 277- | | 2.8.1 1.0.1 | P 1 | 162 | 161 | 191 | 182 | | | |
| 276- | p 4 | 82 | | 163 | 162 | 182 | 183 | | | |

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| 280- | O X Y | FIFLD-1 | FIFLO | ,FIFLD-3, | FIFLD-4 | ,FIELD-5, | FIELDA | FIELE-7 | , F.I.E.C 8, | , rieto. | fIFLN-1,FIFLN-2,FIFLN-3,FIFLN-4,FIFLN-5,FIFLN-6,FIFLE-7,FIFLN-8,FIFLN-9,FIFLN-9,FIFLN-10-10 |
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| DPATO 43 164 163 183 184 DPATO 183 184 184 184 184 184 184 184 184 184 184 184 184 184 184 185 184 185 184 185 184 185 184 184 185 184 185 184 184 185 184 185 184 184 185 184 185 184 185 184 185 184 185 184 185 184 185 184 186 184 185 184 185 184 186 184 185 184 185 184 185 184 185 184 185 184 185 184 185 184 185 184 184 184 184 184 184 184 184 184 184 184 184 184 184 184 184 184 184 <td>280-</td> <td>OPATO</td> <td>282</td> <td>•~:</td> <td>163</td> <td>162</td> <td>182</td> <td>183</td> <td></td> <td></td> <td></td> | 280- | OPATO | 282 | • ~: | 163 | 162 | 182 | 183 | | | |
| DPATQ 183 184 DPATQ 243 184 163 184 DPATQ 243 164 184 185 DPATQ 184 185 164 184 185 DPATQ 3 165 164 184 185 DPATQ 3 165 164 184 185 DPATQ 3 10 1 41 21 DPATQ 3 10 1 41 21 DPATQ 3 10 1 41 21 DPATQ 3 402809 745356 0 942809 745356 0 HXX=10 10 1 1 0 942809 745356 0 HXA=3 16 21 1 0 942809 745356 0 PLOAD 16 1 0 942809 745356 0 PLAD 1 2 1 0 942809 745356 0 PLAD 1 0 1 0 942809 745356 0 PLAD 1 0 0 0 0 0 0 0 PLAD 0 <td>221-</td> <td>DPAT0</td> <td>A 3</td> <td>_</td> <td>164</td> <td>163</td> <td>183</td> <td>184</td> <td></td> <td></td> <td></td> | 221- | DPAT0 | A 3 | _ | 164 | 163 | 183 | 184 | | | |
| CPATO 2M3 164 163 183 184 CPATO 84 165 164 184 185 CPATO 164 184 185 CPATO 165 164 184 185 CPATO 16 1 1 41 21 CPATO 2 10 1 41 21 41 CPATO 3 2 10 1 41 21 41 CATO 3 4 2 1 41 21 41 CATO 3 4 3 4 41 3 44 CATO 4 4 4 4 44 44 44 44 44 CATO 4 | 782- | DPATO | 183 | ~ | 164 | 163 | 183 | 184 | | | |
| CPATO AL 165 164 184 185 CPATO 184 2 165 164 184 185 CPATO 184 3 165 164 184 185 CPATCH 2 1 1 1 1 2 CPATCH 3 P 10 1 1 1 2 CPATCH 4 P 10 1 1 2 CATCH 3 P 10 1 1 2 CATCH 4 P 10 1 1 2 CATCH 5 P 10 1 2 CATCH 6 P 10 1 2 CATCH 7 P 10 1 2 CATCH 8 P 10 1 2 CATCH 9 | 243- | DPATG | E # C | 'n | 164 | 163 | 183 | 184 | | | |
| DPATO 180 2 165 164 189 185 DPATO 280 3 165 164 189 185 DPATO 280 3 165 164 189 185 DPATCH 1 P 10 1 1 2 2 81 DPATCH 2 P 10 1 1 61 41 71 DPATCH 3 P 10 1 1 61 41 71 DPATCH 4 P 10 1 1 61 41 61 MTRX=10 1.0 .942809 .745356 0. 1.0 .942809 .745356 0. PLOAD 3 16 21 1 745356 0. PLOAD 3 16 41 7 -1.814+9 PLOAD 3 16 41 7 -1.814+9 PLOAD 3 16 61 3 -1.814+9 END DATA | 284- | PATO | 30 | | 165 | 104 | 184 | 185 | | | |
| DPATO 284 3 165 164 184 185 DPATCH 1 | 285- | OPATG | 180 | ~ | 165 | 164 | 184 | 185 | | | |
| PATCH 1 P 10 1 1 21 81 P1 P1 P1 P1 P1 P1 P1 P1 P1 P1 P1 P1 P1 | 286- | DPATO | | P 1 | 165 | 164 | 194 | 185 | | | |
| DPATCH 3 P 10 1 1 41 21 41 21 DPATCH 3 P 10 1 1 1 61 41 41 61 41 61 41 61 41 61 41 61 41 61 41 61 41 61 41 61 61 61 61 61 61 61 61 61 61 61 61 61 | 287- | PATCH | | ۵ | 10 | _ | _ | 21 | £ | | |
| DPATCH 3 P 10 1 1 61 41 DPATCH 4 P 10 1 1 6 61 61 MTRX=10 1,0 942809 745356 0, 1,0 942809 745356 0, HMXIO 1.0 942809 745356 0, 1,0 942809 745356 0, PLOAD3 16 21 1 -1,814+9 PLOAD3 16 41 2 -1,814+9 PLOAD3 16 61 3 -1,814+9 PLOAD3 16 61 4 4 -1,814+9 PLOAD3 16 61 4 4 -1,814+9 | 288- | OPATCH | ^ | ۵. | 10 | • | _ | 41 | 21 | | |
| DPATCH 4 P 10 1 1 61 61 61 61 61 MTRX=10 1.0 .942809 .745356 0. +MTRX=10 1.0 .745356 0. +MTRX=10 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1 | 289- | DPATCH | ₩ | Q. | 10 | _ | | 61 | 41 | | |
| MTRX=10 1.0 .942809 .745356 0. 1.0 .942809 .745356 0. +MX10 1.0 .942809 .745356 0. +MX10 1.0 .942809 .745356 0. PLOAD3 16 21814+9814+9 PLOAD3 16 61 3 -1.814+9 PLOAD3 16 61 3 -1.814+9 PLOAD3 16 61 4 -1.814+9 END DATA | -062 | OPATCH | 7 | a . | 10 | - | | 10 | 61 | | |
| +MX10 1.0 .942809 .745356 0. 1.0 .942809 .745356 0. PLNAD3 16 21 1 -1.814+9 PLNAD3 16 41 2 -1.814+9 PLNAD3 16 61 3 -1.814+9 PLNAD3 16 61 4 -1.814+9 END DATA | 291- | MTRX-10 | 1.0 | .942809 | .745356 | • | 1.0 | 942809 | .745356 | • | +MX10 |
| PLOAD3 16 21 1 -1,814+9 PLOAD3 16 41 2 -1,814+9 PLOAD3 16 61 3 -1,814+9 PLOAD3 16 81 4 -1,814+9 END DATA | 292. | OIXX+ | 0. | 942809 | .745356 | • | 0.1 | 942809 | .745356 | • | |
| PLCAD3 16 41 2 PLCAD3 16 61 3 PLCAD3 10 81 4 | 293- | PL CAD3 | | 21 | - | -1.814+9 | • | • | | , | |
| PLFAD3 16 61 3 PLMAD3 16 A1 4 END DATA | -762 | PLOAD3 | 16 | 77 | n: | -1,814+9 | _ | | | | |
| PLMANS 16 A1 4 END DATA | 295- | PLFAn3 | - 4 | ÷. | PT: | *1.814+9 | | | | | |
| END DATA | -962 | PLAAN3 | ٥ | A.1 | 7 | -1.814+9 | | | | | |
| | 297- | END DAT | ⋖ | | | | | | | | |

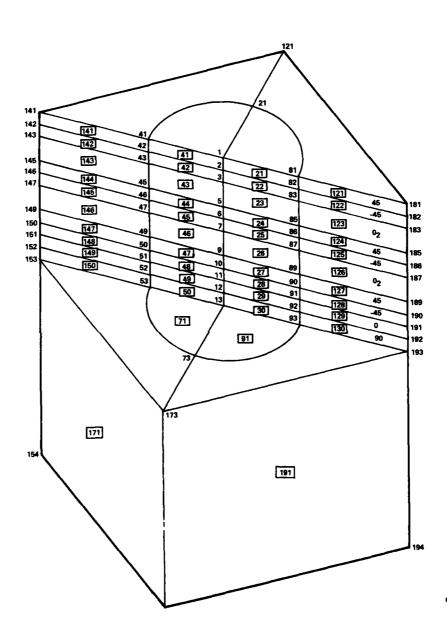
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| C | S DATA DIRECTI | | | c | 0.0 | c | 1259-2-1259 | 259-2-,1259 | 259-2-, 1259 | 259-2-,1259 | 259-2-, 125 | 3560=2=3560 | .1540-2-,3560 | . 1560=2-, 3560 | .3540-2-,3540 | .3540=2=,3540 | -,3560-2 ,3560 | -,3540-2 ,3540 | . 3560=2 ,356 | -,3560-2 ,3560 | -,3560-2 ,3540 | -,3540-2-,3540 | -3560-2-3560 | -,3560-2-,3560 | -,3560-2-,3560 | -,3560-Z-,3560 | 3560-2-,3560 | .3560-2-,3560 | .3560-2-,3560 | .3560-23560- | 3560424,3560 |
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PATCHES-III SCHEMATIC LAMINATE IMPACT IMPACT SITE 12 PLY MODEL

The input data for this model are a direct extension of the 4 ply model with 32 finite elements increased to 88 finite elements. If the findings of the present report are used, then one plane of symmetry (e_1,e_3) can be used to reduce the number of elements. In this instance the topologically equivalent model would have 66 finite elements. To reduce this number further will require a change to the basic layout used in this report.

EMPACT SITE 12 PLY MODEL



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